

Gd³⁺-Ion-Doped Upconversion Nanoprobes: Relaxivity Mechanism Probing and Sensitivity Optimization

Feng Chen, Wenbo Bu,* Shengjian Zhang, Jianan Liu, Wenpei Fan, Liangping Zhou, Weijun Peng, and Jianlin Shi*

Paramagnetic gadolinium (Gd-III)-ion-doped upconversion nanoparticles (UCNPs) are attractive optical-magnetic molecule imaging probes and are a highly promising nanopatform for future theranostic nanomedicine design. However, the related relaxivity mechanism of this contrast agent is still not well understood and no significant breakthrough in relaxivity enhancement has been achieved. Here, the origin and optimization of both the longitudinal (r_1) and transverse (r_2) relaxivities are investigated using models of water soluble core@shell structured Gd³⁺-doped UCNPs. The longitudinal relaxivity enhancement of the nanoprobe is demonstrated to be co-contributed by inner-and outer-sphere mechanisms for ligand-free probes, and mainly by outer-sphere mechanism for silica-shielded probes. The origin of the transverse relaxivity is inferred to be mainly from an outer-sphere mechanism regardless of surface-coating, but with the r_2 values highly related to the surface-state. Key factors that influence the observed relaxivities and r_2/r_1 ratios are investigated in detail and found to be dependent on the thickness of the NaGdF₄ interlayer and the related surface modifications. A two orders of magnitude (105-fold) enhancement in r_1 relaxivity and 18-fold smaller r_2/r_1 ratio compared to the first reported values are achieved, providing a new perspective for magnetic resonance (MR) sensitivity optimization and multimodality biological imaging using Gd³⁺-doped UCNPs.

1. Introduction

Paramagnetic gadolinium (Gd-III)-ion-doped upconversion nanoparticles (UCNPs) have attracted increasing interest as a promising bimodal molecular imaging probe because

of the recent discovery of the magical roles of Gd³⁺ ions, which, after being doped in a UCNP matrix, could result in simultaneous phase/size control and upconversion luminescent efficiency enhancement,^[1] fine-tuning of upconversion emission through the trapping of energy migration,^[2] and more importantly, introduction of a magnetic resonance (MR) contrast imaging modality with well-preserved excellent upconversion luminescence (UCL) capabilities,^[3–11] such as long luminescence lifetime,^[12] improved tissue penetration depth,^[13–15] minimized photodamage,^[16] non-autofluorescence, and non-blinking.^[4,17] Compared with traditional single modal gadolinium chelates, doping Gd³⁺ ions into the crystalline lattice of NaYF₄ enables an extremely high Gd³⁺-payload (over 10⁴, depending on NaGdF₄ thickness) and facilitates a rigid binding of the lanthanide ions, which provides additional and unique advantages for potential MR cellular imaging and, more importantly, reduced risk of toxic ion-leaching *in vivo*.^[18,19]

According to the Solomon-Bloembergen-Morgan (SBM) theory, the paramagnetic relaxation enhancement of gadolinium chelates originates from two mechanisms: the inner-sphere (IS) and outer-sphere (OS) mechanisms.^[20–22] An inner-sphere proton relaxation mechanism involves bonding of water molecules directly to Gd³⁺ ions, while no water to Gd³⁺ ion bonding is needed for an outer-sphere relaxation mechanism. The overall longitudinal relaxivity is the sum of inner-sphere relaxivity, r_1^{IS} , and outer-sphere relaxivity, r_1^{OS} (Supporting Information Section 1). Classically, the number of coordinated water molecules (q) in the first coordination sphere, the mean residence time (τ_m) of water molecule(s), and the rotational correlation time (τ_R) are the three most structure-related parameters that have been actively studied for r_1^{IS} optimization.^[23] The diffusion correlation time (τ_D) is a parameter that highly related to the improvement of r_1^{OS} . Better understanding of the dominant mechanisms and parameters that govern paramagnetic relaxation properties of given probes could lead to a greater enhancement in agent biosafety and MR sensitivity. Generally, for nanoparticle-based contrast agents, it has been well-accepted that 1) loading contrast agents (e.g., Fe₃O₄,^[24] MnO,^[25,26] free Gd³⁺ ions,^[27] or

F. Chen, Prof. W. Bu, J. Liu, W. Fan, Prof. J. Shi
Group of Mesoporous and Low-Dimensional
Nano-materials, State Key Laboratory
of High Performance Ceramics
and Superfine Microstructure
Shanghai Institute of Ceramics
Chinese Academy of Sciences
Shanghai, 200050, People's Republic of China
E-mail: wbbu@mail.sic.ac.cn; jishi@sunm.shcnc.ac.cn
S. Zhang, Prof. L. Zhou, Prof. W. Peng
Department of Radiology
Fudan University Shanghai Cancer Center
Department of Oncology
Shanghai Medical College
Fudan University
Shanghai 200032, China



DOI: 10.1002/adfm.201201469

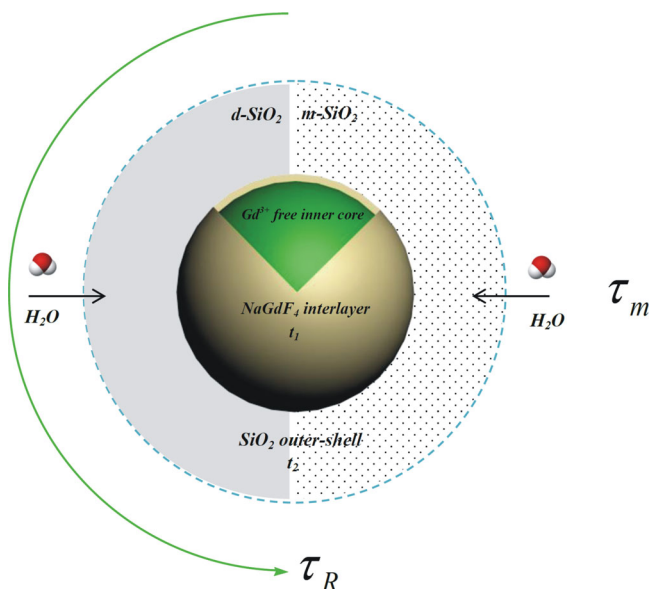
gadolinium chelates^[28] into various matrixes (e.g., silica nanoparticles,^[25,26,29–31] polymersome,^[32] lipid,^[33] fullerene,^[27,28] or viral capsid^[34]) could result in an altered residence time, rotational rate, and finally changed MR relaxivities; and 2) a porous matrix has been commonly reported to be a better choice than a dense one because of the relatively greater accessibility of water molecules.^[25,28,30,32]

Although water soluble Gd³⁺-doped UCNP have been considered an inherently highly promising bimodal imaging nanoplatform,^[7,10] attempts to promote biological imaging applications and future possible medical translation have been hindered largely by their extremely low relaxivity.^[3–9] For example, the first reported longitudinal relaxivity (r_1) value is as low as 0.14 mM^{−1} s^{−1} per Gd³⁺ ion.^[3] Sensitivity optimization efforts have not shown significant breakthrough so far owing to the lack of understanding on the involved relaxivity mechanism. Additionally, most of previous reports have focused only on the T_1 - by ignoring the T_2 -contrast effect, and almost all of the reported Gd³⁺-doped UCNP have been arbitrarily defined as T_1 -contrast agents without addressing the important r_2/r_1 ratios.^[3–10] Here, we aim to focus on following two vital issues of water soluble Gd³⁺-doped UCNP: 1) probing the possible origin of MR longitudinal and transverse contrast enhancements and 2) exploring keys to tailoring the r_2/r_1 ratios and dominant parameters for achieving a remarkably enhanced MR sensitivity.

2. Results and Discussion

We have previously demonstrated that only surface Gd³⁺ ions in the ultrathin NaGdF₄ layer are responsible for shortening the T_1 -relaxation time of water protons and an optimal NaGdF₄ thickness of ca. 1 nm has been suggested for achieving a balanced performance of both UCL and MR modalities.^[7] Here, similar core@shell structured nanocrystals with a Gd³⁺-free NaYF₄:Er/Yb (ca. 22.1 nm) core and a thin NaGdF₄ shell (ca. 1.2 nm) were synthesized (denoted as UCNP_{−1}, Supporting Information Figure S1). As-synthesized hexagonal-phase UCNP_{−1} have no intrinsic aqueous solubility because of the presence of hydrophobic oleate capping ligand.^[7] Shell-thickness tunable and water soluble dense (d-) and mesoporous (m-) silica layers are naturally different from each other in shell-porosity. We suppose the silica coatings could transfer UCNP from the hydrophobic to hydrophilic, and simultaneously alter the aforementioned three critical parameters (q , τ_m , and τ_R), which creates an excellent opportunity to probe the involved relaxivity mechanisms, as well as the dominant parameters that govern the MR sensitivity (Scheme 1). The capability of such silica-coated UCNP for shortening the T_1 and T_2 relaxation time of water protons is reflected by the relaxivities (r_1 and r_2), which are believed to be the result of the complex interplay between structure-related parameters, such as the coordinated-state of surface Gd³⁺ ions, the thickness of NaGdF₄ (t_1), the porosity of protective silica layer, and the corresponding shell thickness (t_2 , Scheme 1).

To demonstrate this, d- and m-SiO₂ coated UCNP_{−1} with tunable shell-thickness were synthesized using previously reported strategies (Figure 1a).^[35,36] The surfactant template in UCNP_{−1}@m-SiO₂ was extracted with a 1 wt% solution of NaCl in



Scheme 1. Well-designed silica protected water soluble core@NaGdF₄ nanoprobe for investigating the MR relaxivity mechanism and sensitivity optimization. The green inner-most core stands for upconversion luminescent Gd³⁺-free NaYF₄:Er/Yb(2/18 mol%) nanocrystals. The ultrathin interlayer NaGdF₄ is the MR contrast signal induced layer (t_1 stands for its thickness), which contains thousands of surface Gd³⁺ ions as payload. Shell thickness (t_2) tunable dense (d-) and mesoporous (m-) silica outer layer are coated over one single particle to create difference in porosity and particle size, which could potentially alter the coordinated number of water molecules (q , surface Gd³⁺ coordinated state dependent), the water residence time (τ_m , shell-porosity dependent), and the rotational correlation time (τ_R , shell-thickness dependent). Note that the sizes of the water molecules are exaggerated for clarity.

methanol.^[30] As-prepared silica coated samples could all be well-dispersed in deionized water without any noticeable aggregation for days. Then, a 3.0 T human magnetic resonance imaging (MRI) scanner was employed for measuring the r_1 and r_2 relaxivities. Contrary to our prediction, for UCNP_{−1}@m-SiO₂, both the r_1 and r_2 relaxivities were virtually independent of the m-SiO₂ shell thickness in a range from 11.1 to 23.4 nm (Figure 1b) and kept nearly unchanged at around 4.8 mM^{−1} s^{−1} and 72.1 mM^{−1} s^{−1}, respectively (Supporting Information Figure S2), while obvious increases in both r_1 and r_2 values were observed in UCNP_{−1}@d-SiO₂ (shell thicknesses ranging from 2.6 to 16.9 nm, Figure 1c, Supporting Information Figure S3). Clearly, both d- and m-SiO₂ shielded UCNP_{−1} show remarkable T_1 and T_2 -MR contrast effects (Supporting Information Figure S2,S3), with average r_2/r_1 ratios estimated to be 12 and 15, respectively (Supporting Information Figure S4), making them more of a T_2 - than a T_1 -contrast agent. Interestingly, only a slight increase in r_1 relaxivity has been observed when both the d- and m-SiO₂ shell thicknesses were approximately 11 nm (5.53 mM^{−1} s^{−1} versus 4.81 mM^{−1} s^{−1}). Surprisingly, the advantage of the d-SiO₂ coating became much more remarkable once the thickness was greater than 15 nm (9.7 mM^{−1} s^{−1} versus 4.94 mM^{−1} s^{−1}). These results suggest that water residence time (shell porosity-dependent) and the rotational correlation time (shell thickness-dependent) should not

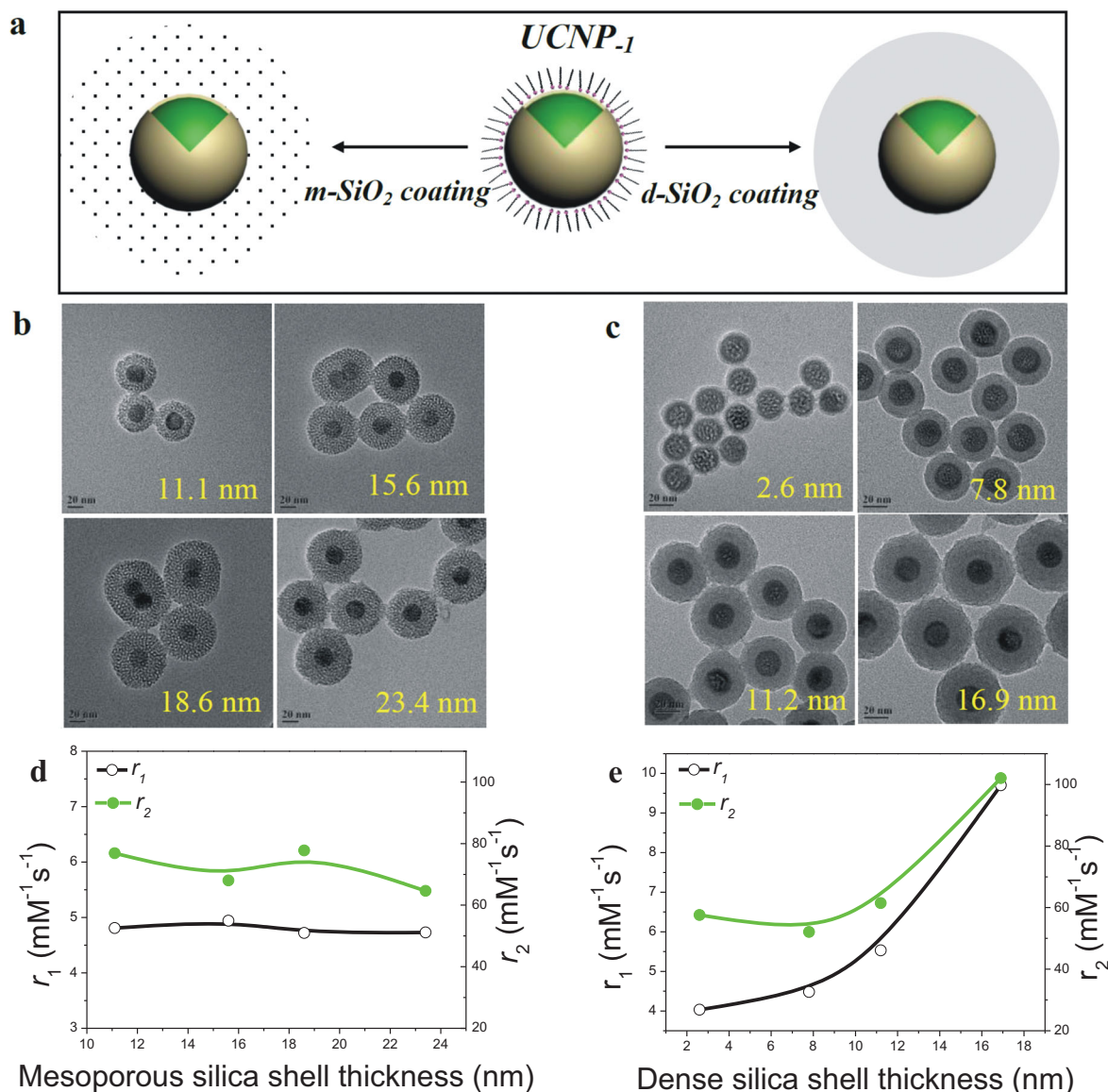


Figure 1. Different MR sensitivity to silica thickness changing in m- and d-silica shielded UCNPs₁. a) Schematic illustration of coating UCNPs₁ with m- and d-SiO₂. b) Transmission electron microscopy (TEM) images of UCNPs₁@m-SiO₂ with thicknesses fixed at 11.1 nm, 15.6 nm, 18.6 nm, and 23.4 nm. c) TEM images of UCNPs₁@d-SiO₂ with thicknesses fixed at 2.6 nm, 7.8 nm, 11.2 nm, and 16.9 nm. Scale bar 20 nm. d) Plot of r_1 and r_2 versus m-SiO₂ shell thickness in (d). e) Plot of r_1 and r_2 versus d-SiO₂ shell thickness in (c). Note: the curves in (d,e) are drawn only as guides for the eyes.

be relaxivity limiting factors for silica shielded probes. If they were, the r_1 value would change significantly with the change of both shell porosity and thickness. This rules out the possible contribution from the inner-sphere mechanism.^[20] Recent reports have already demonstrated that d- and m-SiO₂ shells possess micro- and mesopore channels, respectively, which could enable the passing through of water molecules from the exterior to interior close to the paramagnetic centers.^[25,30,31,37] It is expected that the m-SiO₂ shell, owing to its much larger channels, is more permeable to water molecules and more useful for relaxivity enhancement.^[25,30] Our observations in silica shielded UCNPs₁, however, do not follow the expectation and may indicate a quite different MR relaxivity mechanism.

To gain deeper insight into the dynamic processes involved between nanoprobes and water molecules, nuclear magnetic relaxation dispersion (NMRD) profiles of UCNPs₁@d-SiO₂ (shell thicknesses of 2.6 and 16.9 nm) and UCNPs₁@m-SiO₂ (thickness: 15.6 nm) were tested over a wide range of Larmor frequencies ranging from 0.01 to 35 MHz at 310 K. To the best of our knowledge, this is the first attempt to study the field-dependent relaxivity of water-soluble Gd³⁺-doped UCNPs. Results show that all three profiles have a similar shape, with one relaxivity bump located at 20 MHz (UCNPs₁@d-SiO₂ (2.6 nm), Figure 2a) and the other two shifting toward higher frequency around 30 MHz (Figure 2b,c). Such relaxivity peaks are usually observed for gadolinium chelates with slow molecular motion and are

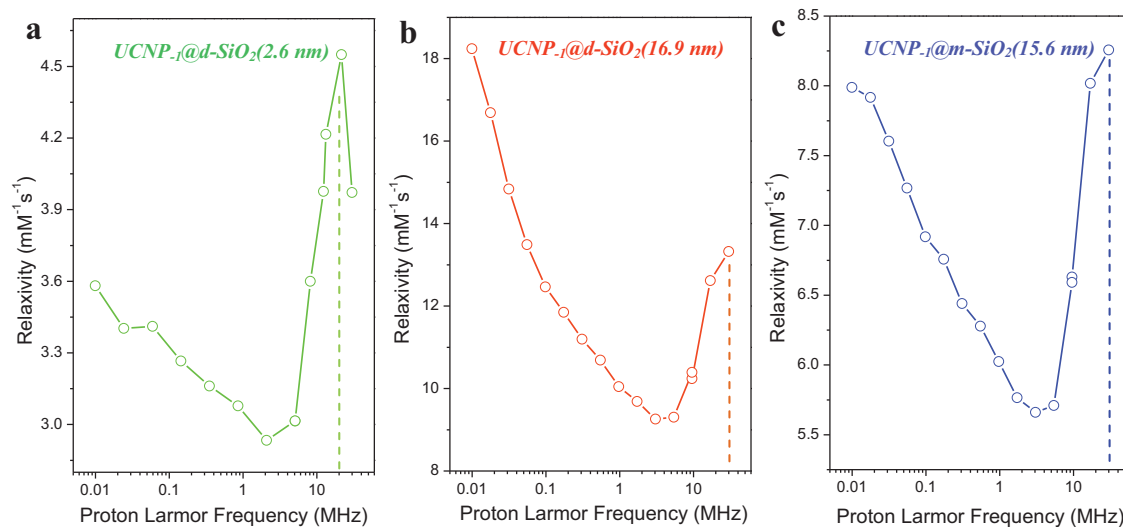


Figure 2. $1/T_1$ NMRD profiles of silica-shielded UCNP. a) UCNP₁@d-SiO₂(2.6 nm), b) UCNP₁@d-SiO₂(16.9 nm), and c) UCNP₁@m-SiO₂(15.6 nm). Samples were tested with Larmor frequencies ranging from 0.01 to 30 MHz at 310 K.

indicative of the presence of a slow tumbling system that the water molecules interacting with.^[20] Quick decrease rates observed between 0.01 and 1 MHz in all three profiles are quite abnormal compared with those of gadolinium chelates, making the NMRD profile fit using a standard model inapplicable.^[20]

Since the direct bonding of surrounding water molecules to the surface Gd³⁺ ions, which are located at the interlayer in silica shielded UCNP₁, might not be possible, we propose that the observed longitudinal relaxivity (r_1) originates mainly from an outer-sphere mechanism caused by the translational diffusion of water molecules penetrating through the water permeable silica shell. Usually the higher the shell porosity, the greater the mobility of diffusion outer-sphere water molecules will be, leading to a decreased correlation diffusion time and finally reduced r_1^{OS} relaxivity.^[20,28] With the presence of ca. 2 nm sized mesoporous channels in the UCNP₁@m-SiO₂,^[38] high and nearly unaffected water mobility would be guaranteed in pore channels, even at the increased m-SiO₂ shell thickness, which explains the insensitivity of r_1 -relaxivity to the m-SiO₂ shell-thickness in the case of UCNP₁@m-SiO₂ (Figure 1d). In stark contrast, the mobility of the outer-sphere water diffusion in the dense silica shell would logically be more restricted than in mesopore channels, especially at the increased shell thickness, which results in increased correlation diffusion time and the increased r_1^{OS} relaxivity (Figure 1e).^[28] This inference is confirmed by the fact that UCNP₁@m-SiO₂ with template CTAB (cetyl trimethylammonium bromide) kept in the mesopore channels has a higher r_1 relaxivity (7.17 mM⁻¹ s⁻¹, Supporting Information Figure S5) than the sample with the template being extracted (Figure 1d, average r_1 = 4.8 mM⁻¹ s⁻¹).

Next, we speculated that although both of the d- and m-SiO₂ shell could allow the diffusion of water molecules in and out, the strong coordination between adjacent silanol groups and surface Gd³⁺ ions may be very likely to prevent water molecules from direct coordinating to the surface Gd³⁺ ions, leading to the “quenching” of the possible contribution from an inner-sphere mechanism. To validate this, we did not coat any silica

shell on UCNP and, instead, removed the oleate capping ligand via a modified acid-treatment method to create a ligand-free UCNP₁ (Figure 3a, Supporting Information Figure S6). Direct bonding of water molecules with surface lanthanide ions in ligand-free UCNP is entirely unaffected.^[39] As expected, we found a much higher r_1 value of 10.4 mM⁻¹ s⁻¹ using the same relaxivity testing methodology (Figure 3b-left). In comparison with the r_1 relaxivity of UCNP₁ coated with an ultrathin d-SiO₂ (shell thickness ca. 2.6 nm, r_1 values 4.03 mM⁻¹ s⁻¹, Supporting Information Figure S3a-c), the contribution from the co-existing inner-sphere mechanism can be verified. However, the attempt to ascertain the possible number of coordinated water molecules in ligand-free UCNP₁ is highly challenging and the validity of defining such a value using traditional methods (most of which are only suitable for estimating q value of “soft” organic gadolinium chelates but not “rigid” inorganic nanocrystals)^[40–42] is still very controversial. We also noticed that, compared to UCNP₁@d-SiO₂ (shell thickness ca. 2.6 nm, r_2 values 57.58 mM⁻¹ s⁻¹, Supporting Information Figure S3a-c), the r_2 relaxivity of ligand-free UCNP₁ does not show the same remarkable increase as the r_1 value with its value estimated to be 63.16 mM⁻¹ s⁻¹ (Figure 3b-right), suggesting the absence of the inner-sphere and the dominant contribution from outer-sphere mechanism during T_2 -contrast imaging. Moreover, the r_2/r_1 ratio of ligand-free UCNP₁ is found to be 6.07 (Figure 3b), which is only half that of silica shielded samples (average r_2/r_1 = 12 to 15, Supporting Information Figure S4).

Clinically, a T_1 -contrast agent (r_2/r_1 ratio close to 1) with high relaxivity is much more desirable than T_2 -agent (r_2/r_1 > 10). For example, an intrinsic dark signal in T_2 -weighted MRI may mislead the clinical diagnosis because lesions and tumors labeled with T_2 -contrast agents can be confused with other hypointense areas such as bleeding, calcification, or metal deposition. The keys to tailoring the r_2/r_1 ratio of Gd³⁺-doped UCNP are still not yet well-documented. We have shown that the silica shell coating usually leads to a T_2 -contrast agent (Supporting Information Figure S4). Here, a larger Gd³⁺-free core (>30 nm), an

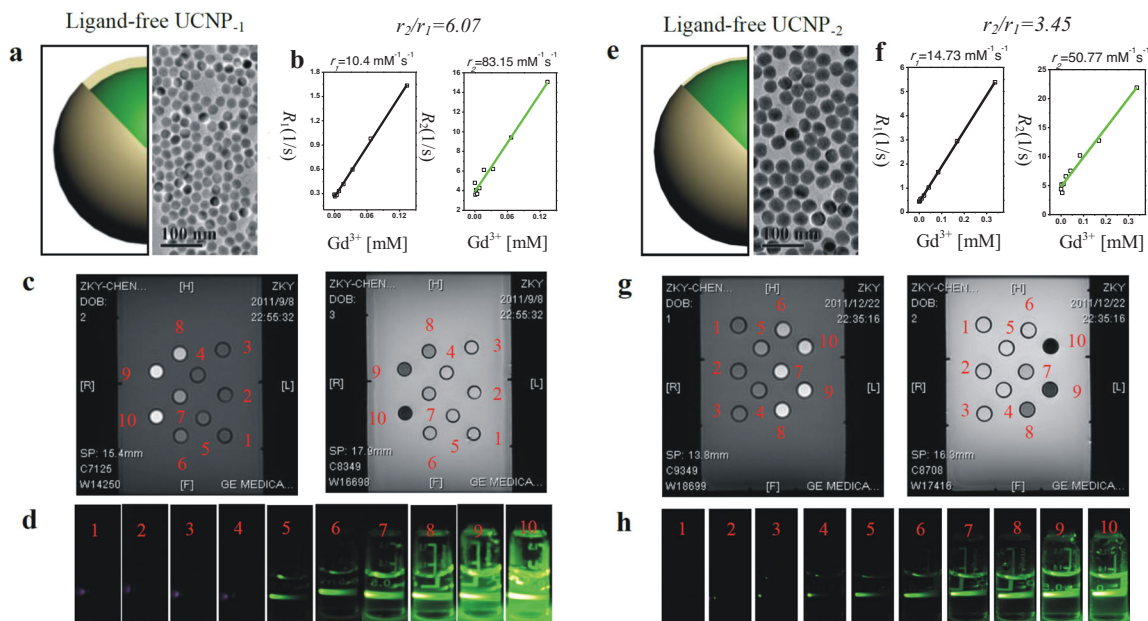


Figure 3. Achieving a small r_2/r_1 ratio by decreasing thickness of NaGdF₄ in ligand-free UCNPs. a) Schematic illustration (left) and TEM image (right) of ligand-free UCNP₁ (NaYF₄:Er/Yb core size \approx 22.1 nm; NaGdF₄ thickness \approx 1.2 nm). b) Corresponding plots of R_1 (left) and R_2 (right) versus Gd³⁺ ions concentrations. c) Corresponding T_1 - (left) and T_2 -maps (right). d) Digital images of samples under excitation of NIR laser (λ = 980 nm, power = 2.0 W). e) Schematic illustration (left) and TEM image (right) of ligand-free UCNP₂ (NaYF₄:Er/Yb core size \approx 33.4 nm; NaGdF₄ thickness \approx 0.9 nm). f) Corresponding plots of R_1 (left) and R_2 (right) versus Gd³⁺ ions concentrations. g) Corresponding T_1 - (left) and T_2 -maps (right). h) Digital images of samples under excitation of NIR laser (λ = 980 nm, power = 2.0 W). The slopes in (b) and (f) indicate the r_1 or r_2 relaxivities.

ultrathin NaGdF₄ outer shell (<1 nm), and a ligand-free water soluble surface may be important elements for achieving a Gd³⁺-doped UCNP-based T_1 -contrast (r_2/r_1 ratio significant lower than 10). We have demonstrated this by synthesizing new NaYF₄:Er/Yb@NaGdF₄ nanoparticles (denoted as UCNP₂) with the diameter of NaYF₄:Er/Yb core fixed at 33.4 nm and NaGdF₄ outer shell thicknesses fixed at ca. 0.9 nm (Figure 4a). The thickness (or the distribution) of the NaGdF₄ layer over the NaYF₄ core has been found to be a key in influencing the observed MR relaxivity,^[7,8] however, it is still a great challenge for one to ascertain the exact shell thickness (or distribution) of the NaGdF₄ layer due to the very close crystalline structures and compositions in core@shell structured UCNPs.^[2,43] All the thicknesses of NaGdF₄ in this study have been estimated using our previously reported method.^[7] The MR relaxivity estimation shows a sharp increase of r_1 value from 10.4 mM⁻¹ s⁻¹ up to 14.73 mM⁻¹ s⁻¹ (Figure 3b-left and 3f-left), when the NaGdF₄ thickness decreases from ca. 1.2 nm to ca. 0.9 nm, in agreement with a previous report.^[7] In comparison with the first reported r_1 value of 0.14 mM⁻¹ s⁻¹,^[3] a two orders of magnitude (105-fold) enhancement in r_1 relaxivity has been achieved in this case. Surprisingly, we also succeeded in further decreasing the r_2/r_1 value to as low as 3.45 in ligand-free UCNP₂, which is 18-fold lower than the first reported ratio (r_2/r_1 = 62.14)^[3] and nearly 2-fold smaller than the extremely small-sized iron oxide nanoparticles (ESION, size = 3 nm, r_2/r_1 = 6.12).^[44] The latter has recently been demonstrated to be an efficient T_1 -contrast agent for imaging blood pool.^[44] Further calculation shows that ligand-free UCNP₂ possesses an extremely high Gd³⁺ ions payload (approximately 1.55×10^4) with particle

relaxivities estimated to be $r_1 = 2.28 \times 10^5$ mM⁻¹ s⁻¹ and $r_2 = 7.87 \times 10^5$ mM⁻¹ s⁻¹ (Supporting Information Table S1), holding great potential for future UCL/MR based biological imaging, such as stem cell labeling and tracking in vivo.

Furthermore, we provide the evidence that extremely thick d-SiO₂ coating might represent a simple but efficient way to create an excellent T_2 -weighted MR contrast agent (r_2/r_1 ratios >20). Using UCNP₂@d-SiO₂ (10.8 nm) as seeds (Figure 4a), another two dense silica shielded UCNPs with shell thicknesses estimated to be 19.5 and 35.3 nm, respectively, were prepared via a well-established re-growth technique (Figure 4e,i). Relaxivity study showed an obvious decrease in r_1 from 14.73 (ligand-free UCNP₂, Figure 3b-left) to 9.29 mM⁻¹ s⁻¹ (UCNP₂@d-SiO₂ (10.8 nm), Figure 4d-left), demonstrating the “quenching” of the inner-sphere contribution after silica-shielding once again. The r_1 relaxivity shows an expected increase from 9.29 to 12.38 mM⁻¹ s⁻¹ (Figure 4d-left and h-left) as the increase of the d-SiO₂ shell from 10.8 to 19.5 nm, however, suffers from a quick decrease back to 9.52 mM⁻¹ s⁻¹ once the silica shell thickness increases to 35.3 nm (Figure 4l-left), suggesting an optimal shell thickness for r_1 -relaxivity might be around 20 nm. The corresponding r_2 values experience a continuous increase from 88.68 to 193.05 mM⁻¹ s⁻¹ (Figure 4d-right, h-right and l-right), resulting in a remarkably enhanced r_2/r_1 ratio from 9.55 to 20.3.

Based on our explanation of the relaxivity mechanism, to give an in vivo UCL/MR bimodal imaging demonstration, nanocrystals with Er³⁺/Tm³⁺/Yb³⁺ co-doped NaYF₄ cores and ultrathin NaGdF₄ shells (NaYF₄:Er/Tm/Yb@NaGdF₄, denoted as UCNP₃, Supporting Information Figure S9a) were designed. The successful co-doping of Er³⁺/Tm³⁺/Yb³⁺ ions has been confirmed

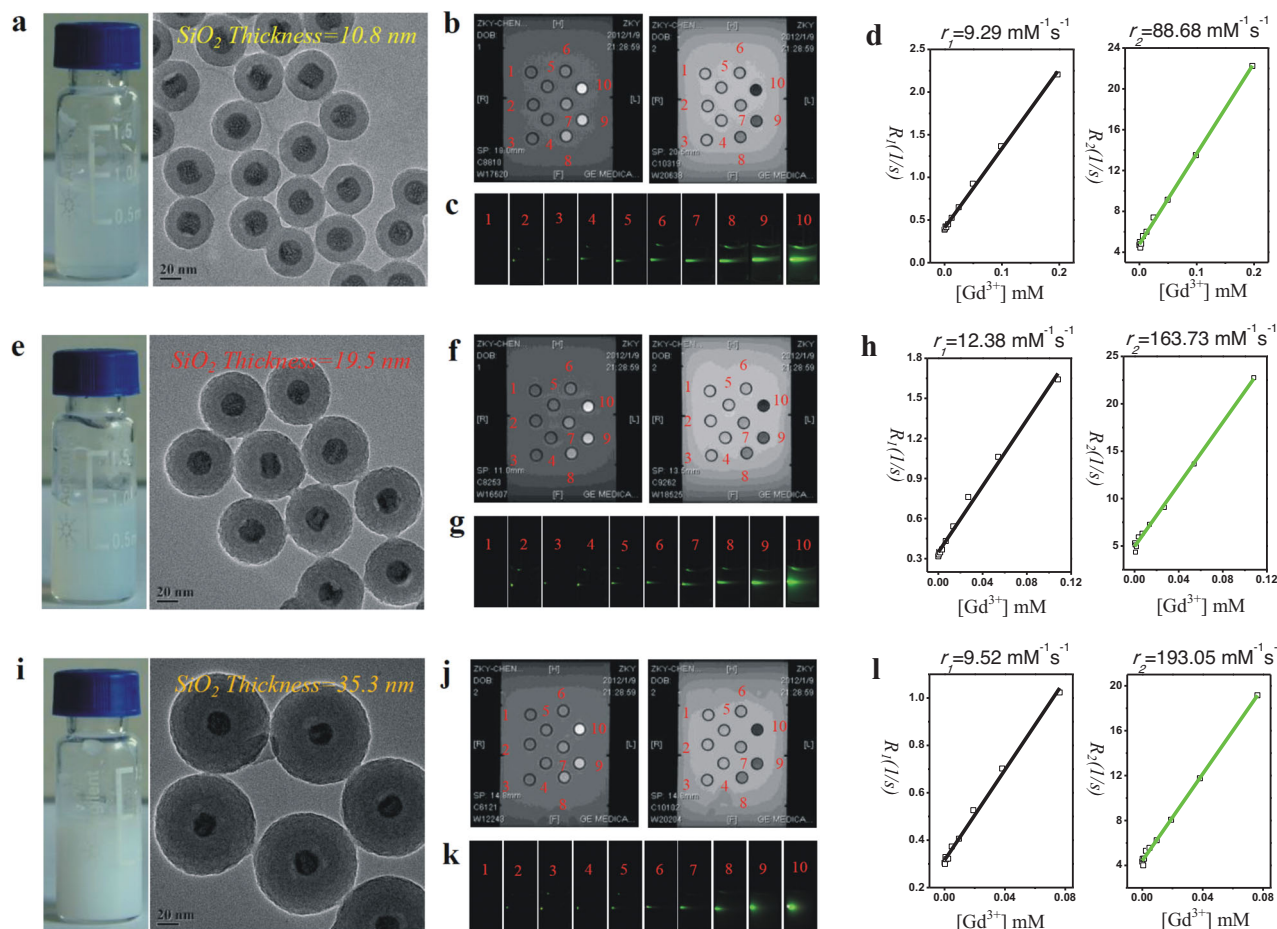


Figure 4. Probing the optimal d-SiO₂ shell thickness and high r_2/r_1 ratio in water soluble UCNPs.₂@d-SiO₂. a) Digital image (left) and TEM image of UCNPs.₂@d-SiO₂ (10.8 nm). b) T₁- and T₂-maps of UCNPs.₂@d-SiO₂ (10.8 nm) with varied Gd³⁺ concentrations. c) Digital images of samples under excitation of NIR laser (λ = 980 nm, power = 2.0 W). d) Plots of R_1 (left) and R_2 (right) versus Gd³⁺ ions concentrations. e) Digital image (left) and TEM image of UCNPs.₂@d-SiO₂ (19.5 nm). f) T₁- and T₂-maps of UCNPs.₂@d-SiO₂ (19.5 nm) with varied Gd³⁺ concentrations. g) Digital images of samples under excitation of a NIR laser (λ = 980 nm, power = 2.0 W). h) Plots of R_1 (left) and R_2 (right) versus Gd³⁺ ions concentrations. i) Digital image (left) and TEM image of UCNPs.₂@d-SiO₂ (35.3 nm). j) T₁- and T₂-maps of UCNPs.₂@d-SiO₂ (35.3 nm) with varied Gd³⁺ concentrations. k) Digital images of samples under excitation of a NIR laser (λ = 980 nm, power = 2.0 W). l) Plots of R_1 (left) and R_2 (right) versus Gd³⁺ ions concentrations. The slopes indicate the r_1 or r_2 relaxivities.

by the emission bands from ultraviolet (UV, ≈ 350 to 370 nm ($^1D_2 \rightarrow ^3H_6$)), visible (Vis, blue band ≈ 460 to 500 nm ($^1G_4 \rightarrow ^3H_6$); green band ≈ 530 to 570 nm ($^4S_{3/2} \rightarrow ^4I_{15/2}$)), and red band ≈ 630 to 700 nm ($^4F_{9/2} \rightarrow ^4I_{15/2}$)), and near-infrared (NIR, ≈ 770 to 810 nm ($^1G_4 \rightarrow ^3F_4$)) regions under the NIR laser (λ = 980 nm) excitation (Supporting Information Figure S9b and Table S2), enabling the NIR-to-vis and NIR-to-NIR small animal imaging in vivo. Ligand-free UCNPs.₃ ($r_1 = 12.5$ mM⁻¹ s⁻¹, $r_2 = 92.6$ mM⁻¹ s⁻¹, $r_2/r_1 = 7.7$, Supporting Information Figure S10) was intravenously administrated (dose: 4.8 μ mol Gd/kg) in SD mouse (≈ 170 g) and imaged in a 3.0 T MRI scanner. Clinically used Magnevist (Gd-DTPA, DTPA: diethylenetriaminepentaacetate, Supporting Information Figure S11a) and ligand-free 8 nm sized superparamagnetic iron oxide nanoparticles (SPIONs, Supporting Information Figure S11d) were also used as control groups. Changes in the MR signal intensity (SI) in liver, one of the main probe-uptake organs,^[18,45] were monitored

at timed intervals, which shows clearly enhanced (+37.9%) and reduced SI (-38.1%) values of the T₁- and T₂-models, respectively, after 30 min (Figure 5c,d), demonstrating the potential of the probe as a dual-weighted MR contrast agent in vivo. In contrast, Gd-DTPA and SPION could only work better in either a T₁- or T₂-model (Supporting Information Figure S11a-c and d-f). For in vivo UCL imaging, a dose of 67.4 μ mol Gd/kg of probe was administrated into Kunming mouse (≈ 30 g) through tail vein injection. In vivo imaging was performed using a home-made UCL in vivo imaging system.^[46] Excitation was provided by a continuous wave (CW) infrared laser at 980 nm and UCL signals were collected at 800 ± 12 nm and analyzed with Kodak Molecular Imaging Software. After 30 min, the whole-body UCL imaging was performed and the results show a clear signal from liver with no autofluorescence (Figure 5c, in vivo). Rapid accumulation of nanoparticles in liver and spleen have also been reported previously.^[18,47] In situ and ex vivo UCL

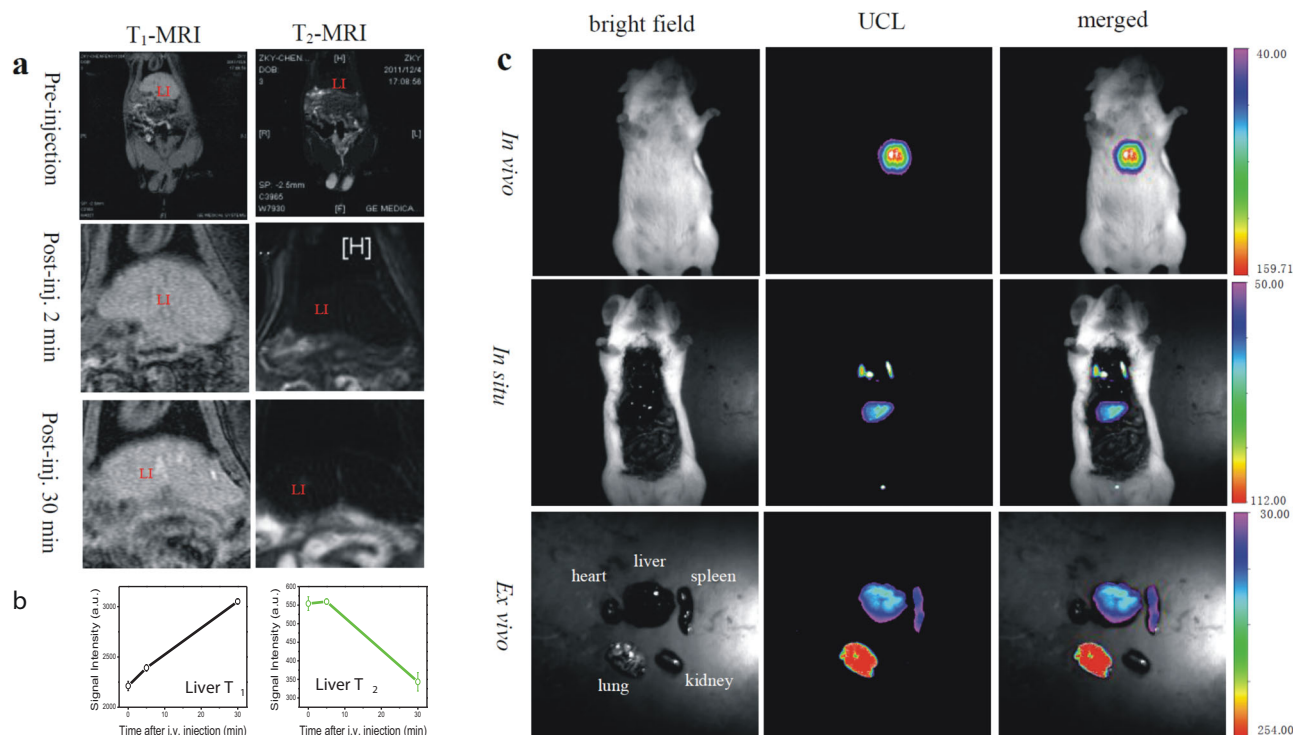


Figure 5. In vivo UCL/MR bimodal imaging using water-soluble ligand-free UCNP₃. a) In vivo MR T₁- and T₂-weighted images of SD mouse. Images in mouse's liver (LI) were monitored by different time points (pre-injection, post-injection 2 min, and post-injection 30 min) after i.v. injection (dose: 4.8 μ mol Gd/kg). b) Plots of time-dependent signal intensity of the liver in T₁-model (left) and T₂-model (right). c) In vivo, in situ, and ex vivo upconversion luminescence imaging at 30 min post-injection of ligand-free probe-4 (dose: 67.4 μ mol Gd/kg).

images further show the uptake of the probe by the spleen and lung (Figure 5c, in situ and ex vivo). These results demonstrate the great potential of the probe as an optical/magnetic dual-modal imaging agent in vivo.

With the increasing attention being focused on designing inorganic-based nanoparticulate contrast agents,^[48] Gd³⁺-ion-doped (or containing) nanocrystals might hold great potential to solve the problems of traditional gadolinium chelates, such as single modality, low sensitivity, lack of specific targeting, possible renal dysfunction, and others.^[49,50] Although previous reports have already demonstrated the capability of MR contrast enhancement effect in Gd³⁺ ions containing nanocrystals,^[3–10, 51–53] the reported longitudinal and transverse relaxivities have shown great variations with no reasonable explanations being provided. Probing the involved relaxivity mechanism remains an open challenge and could potentially be solved using systematic relaxivity investigations of well-selected models. Based on our previous successful demonstration of the dominant T₁-relaxation time shortening effect of surface Gd³⁺ ions in core@shell structured upconversion nanocrystal,^[7] we attempt to address these problems by selecting d- and m-SiO₂ shielded NaYF₄:Er/Yb@NaGdF₄ (Scheme 1) as models by considering that silica coating is water soluble and permeable, highly biocompatible, and the porosity and thickness are easily controllable.^[35,36] Using these models, contrary to our prediction, the mesoporous silica shell with higher porosity has been shown to not be very helpful for relaxivity enhancement, while

due to the dominant role of outer-sphere mechanism in silica-shielded probe, the dense silica shell has been proven to be much more useful not only for a greater relaxivity enhancement, but also potentially for a better protection of nanoprobe.

The third model, ligand-free probe, has turned out to be even more useful for mechanism probing. Because of the metal coordination between surface lanthanide ions and carboxylate,^[54] the well-documented advantages of oleic acid in thermal decomposition synthesis of Gd³⁺-doped UCNP are also the main cause of intrinsic aqueous non-solubility and the origin of no direct water-to-Gd³⁺ coordination. Direct bonding of surrounding water molecules to the surface Gd³⁺ ions would be impossible if the oleate ligand remained capped on the crystal surface or was replaced by other strongly coordinated ligands, such as silanol groups. We have confirmed this inference by comparing the relaxivity of silica-shielded and ligand-free probes, leading us to the conclusion that, as is the case for Gd-DTPA (number of coordinated water molecules is one), both the inner- and out-sphere mechanisms do co-exist in ligand-free Gd³⁺-doped UCNP. This finding provides useful guidance for surface modification and further sensitivity optimization of Gd³⁺-ion-doped inorganic nanocrystals. Additionally, suitable techniques for ascertaining the number of coordinated water molecules in ligand-free probes are so far unavailable, however, due to the possible difference in coordinated states of thousands of surface Gd³⁺ ions, not an exact *q* value but possibly a *q* value distribution is more practical. Moreover, the abnormally quick decrease

rates in NMRD profiles between 0.01 and 1 MHz suggest the inapplicability of standard models for Gd³⁺-doped nanocrystals during profile-fitting, highlighting the necessity for mathematic model modifications in upcoming research.

3. Conclusions

In conclusion, in an attempt to understand the relaxivity mechanism of water-soluble core@shell structured Gd³⁺-doped UCNP, systematic investigations on well-selected models have been carried out. Inner- and outer-sphere r_1 -relaxivity mechanisms have been demonstrated to be coexisting in ligand-free probes, while the outer-sphere mechanism has been suggested to be the main contribution to that of silica-shielded probes. The origin of r_2 -relaxivity is inferred mainly from an outer-sphere mechanism with the r_2 value being surface-state related. Key factors for tuning r_2/r_1 ratios (ranging from ≈ 3 to ≈ 20 , Supporting Information Table S3) have been found to be highly dependent on the thickness of NaGdF₄ interlayer and surface modifications, with the lowest ratio achieved equal to 3.45. An in vivo UCL/MR bimodal imaging using nanoprobe designed based on our mechanism probing has also been demonstrated. Our new findings provide not only deeper insight into the origin of contrast enhancements of water soluble Gd³⁺-doped UCNP, but also useful strategies for achieving both high MR relaxivities and tunable r_2/r_1 ratios. These, we believe, will be highly favorable for enhancing and optimizing the sensitivity of other Gd³⁺-ion-containing nanoparticulate probes (such as Gd₂O₃, GdPO₄, and GdF₃)^[51–53] and, at the same time, speeding up the biological applications and possible medical translation of these promising optical-magnetic bimodal molecule imaging probes.

4. Experimental Section

Synthesis of UCNP₁: First, Gd-free core, NaYF₄:Er(2%)/Yb(18%), was prepared. YCl₃·6H₂O (970.75 mg, 3.2 mmol), YbCl₃·6H₂O (279 mg, 0.72 mmol), ErCl₃·6H₂O (30.54 mg, 0.08 mmol) in deionized water (8 mL) were added to a 100 mL flask containing oleic acid (30 mL) and 1-octadecene (60 mL). The solution was stirred at room temperature for 1 h. The mixture was slowly heated to 120 °C under an argon atmosphere to remove water and was maintained at 160 °C for about 1 h until a homogeneous transparent yellow solution was obtained. The system was then cooled to room temperature with the flowing of argon. Then 10 mL a methanol solution of NH₄F (592.6 mg, 16 mmol) and NaOH (400 mg, 10 mmol) was added and the solution was stirred at room temperature for 2 h. After the methanol evaporated, the solution was heated to ≈ 280 to 290 °C and kept for 1.5 h, then cooled to room temperature. The resulting nanoparticles were precipitated by the addition of 40 mL ethanol and collected using centrifugation at 10 000 r/min for 10 min. The product was redispersed with 10 mL cyclohexane and precipitated by adding 30 mL ethanol, then collected by the same centrifugation. After four times washing, the final product was redispersed in 40 mL cyclohexane.

An ultrathin NaGdF₄ layer was grown over the Gd-free core using a seed-mediated method. GdCl₃·6H₂O (11.07 mg, 0.042 mmol) in deionized water (0.5 mL) were added to a 100 mL flask containing oleic acid (15 mL) and 1-octadecene (30 mL). The solution was stirred at room temperature for 1 h. Then the mixture was slowly heated to 120 °C under an argon atmosphere to remove water and then maintained at 160 °C for about 1 h until a homogeneous transparent yellow solution

was obtained. The system was cooled to room temperature with the flowing of argon. Then, 5 mL pre-prepared Gd-free core (dispersed in cyclohexane) was added and kept for another 30 min before heating to 80 °C to remove cyclohexane. Next, 1.25 mL methanol solution of NH₄F (6.241 mg, 0.169 mmol) and NaOH (4.212 mg, 0.105 mmol) was added and the solution was stirred at room temperature for 2 h. After the methanol evaporated, the solution was heated to ≈ 270 to 280 °C and kept for 1.5 h, before cooling to room temperature. The resulting nanoparticles were precipitated by the addition of 20 mL ethanol and then collected by centrifugation at 10 000 r/min for 10 min. The product was redispersed with 5 mL cyclohexane and precipitated by adding 15 mL ethanol, then collected by the same centrifugation. After washing four times, the final product was redispersed in 10 mL cyclohexane.

The procedures for the synthesis of UCNP₂ and UCNP₃ were similar to those described above, except different amounts of NaGdF₄ precursors (such as GdCl₃·6H₂O, NH₄F, and NaOH) or new lanthanide ions (such as TmCl₃) were added.

Synthesis of UCNP₁@d-SiO₂: In a typical synthesis of UCNP₁@d-SiO₂, Igepal CO-520 (NP-5, 1 mL) was dispersed in cyclohexane (20 mL) in a 50 mL three-necked flask and stirred for 40 min. Then, oleic acid capped UCNP₁ in cyclohexane solution (1.5 mL, ≈ 100 mM) was added into the cyclohexane/NP-5 mixture and stirred for 3 h. Then ammonia (140 μ L, 30%) was added dropwise, and the system was sealed and stirred for another 2 h. 200 μ L tetraethyl orthosilicate (TEOS) was delivered into the system at a rate of 200 μ L h⁻¹ using a syringe pump (WZS-50F6). The mixture was sealed and kept under magnetic stirring for 36 h at ≈ 18 to 22 °C before adding methanol to collect particles. The product was precipitated with excess hexane and collected by centrifugation. Then the particles were redispersed in ethanol under ultrasonic treatment, precipitated with excess hexane, and collected by centrifugation. The as-synthesized particles were washed three times using this procedure to remove excess NP-5. The as-obtained nanoparticles could be dispersed in ethanol or deionized water. For the synthesis of UCNP₁@d-SiO₂ with tunable shell thickness, a different amount of TEOS was used.

Synthesis of UCNP₁@m-SiO₂: UCNP₁@CTAB was prepared. In a typical procedure, the UCNP₁ in chloroform (3 mL) was added into aqueous cetyl trimethylammonium bromide (CTAB) solution (120 mL, 0.1 M), and the resulting solution was sonicated using a probe-type sonicator at 600 W for 30 min to form a transparent UCNP₁@CTAB solution. The excess amount of CTAB could be removed by decreasing the temperature to 0 °C.

The resulting solution (30 mL) was added to a mixture of water (60 mL), ethylacetate (4 mL), and NH₄OH solution (600 μ L). The mixture was sonicated for 2 h with a 53 kHz frequency. Then TEOS (100 μ L) was added dropwise and the then mixture was reacted for another 10 h. As-synthesized UCNP₁@m-SiO₂ was collected by centrifugation and washed three times with ethanol to remove the unreacted species. The surfactant template was then extracted with a 1 wt% solution of NaCl in methanol. After that, the product was washed with distilled water and ethanol before being redispersed in water.

Synthesis of Ligand-Free UCNP₁: In a typical synthesis, deionized water (5 mL) was added in a bottle followed by carefully adding cyclohexane-dispersed UCNP₁ (5 mL). After that, HCl (20 μ L, 1.2 M) was added before shaking the bottle for 1–2 min. Nearly all of the hydrophobic oleate-capped UCNP₁ could be transferred to the water phase after shaking. Then, the water phase was quickly collected using separating funnel and centrifuged (13 000 r/min) for 10 min. The sample was then washed three times with acetone, resulting in ligand-free UCNP₁. As-synthesized product could be well-dispersed in deionized water.

NMRD Profiles: The 1/T₁ NMRD profiles were measured at a number of relaxation filed values in the 0.01–35 MHz range, using the Fast Field Cycling NMR Relaxometer produced by Stelar S.R.L. Standard pre-polarized (PP/S) and non-polarized (NP/s) acquisition sequences were used with the following parameters: temperature = 37 °C, relaxation field = 0.01–35 MHz, acquisition field = 16.3 MHz, nucleus = ¹H, spectrometer frequency = 16.3 MHz, number of scans = 1, and 90° pulse width = 8.1 μ s.

In Vivo MR Imaging: In vivo MR imaging of SD mouse (≈ 170 g) using Gd-DTPA (dose: 294 μ mol Gd/kg), 8 nm sized ligand-free SPION (dose:

18.3 $\mu\text{mol Fe/kg}$), and ligand-free UCNP₃ (dose: 4.8 $\mu\text{mol Gd/kg}$) as a contrast agent was performed using a 3.0 T clinical MRI instrument. After tail vein injection, the mouse was taken for the T_1 - and T_2 -weighted MRI tests. The SPGR sequence was used for T_1 -weighted MRI with the following parameters: TR/TE = 85/3.5 ms, flip angle: 80, field of view (FOV): 15 cm; matrix: 384×192 , number of excitations (NEX): 3; slice thickness = 2 mm; space = 1 mm; and coil: QUADKNEE. The FRFSE sequence was used for T_2 -weighted MRI test with the following parameters: TR/TE = 3720/105 ms, ETL: 16, FOV: 12 cm; matrix: 256×192 , number of excitations (NEX): 2; slice thickness = 2 mm; space = 1 mm; and coil: QUADKNEE. All the images were then analyzed using the workstation provided by GE healthcare. Signal intensities were estimated using the eFilm Workstation.

In Vivo Upconversion Luminescence (UCL) Imaging: In vivo and ex-vivo UCL imaging was performed using a home-made UCL in vivo imaging system from Prof. Li's group (Advanced Materials Laboratory, Department of Chemistry, Fudan University). Images of UCL signals were analyzed with Kodak Molecular Imaging Software. Excitation was provided by the continuous wave (CW) infrared laser at 980 nm and UCL signals were collected at 800 ± 12 nm. A Kunming mouse (≈ 30 g) was anesthetized and tail vein injected with ligand-free UCNP-3 (dose: 67.4 $\mu\text{mol Gd/kg}$). After 30 min post-injection, the whole-body UCL imaging was performed. After that, the major organs were removed for ex vivo imaging.

Other Characterizations: TEM images were recorded on a JEOL 200CX microscope with an accelerating voltage of 200 kV. Standard TEM samples were prepared by dropping dilute products onto carbon-coated copper grids. The RE^{3+} concentrations of samples were measured by inductively coupled plasma optical emission spectrometry (ICP-OES). Upconversion luminescence emission spectra were recorded on FluoroLog-3 Spectrofluorometer (Jobin Yvon, France), with the excitation of a 450 W xenon lamp and an external ≈ 0 to 1 W adjustable 980 nm semiconductor laser (Beijing Hi-tech Optoelectronic Co., China). The MR imaging experiments in water solutions were performed on a 3.0-T clinical MRI instrument (GE Signa 3.0T) and the pulse sequence used was a T_1 -weighted FSE-XL/90 sequence with the following parameters: TR/TE = 1000, 2000, 3000, 4000/7.9 ms; FOV: 18 cm; matrix: 128×128 ; NEX: 2; slice thickness = 2 mm; space = 0.5 mm; FOV: 18 cm; and coil: QUADKNEE. Parameters for the T_2 -weighted fast-recovery fast spin-echo (FR-FSE) sequence were as follows: TR = 4000 ms, slice thickness = 3.0, TE = 98 ms, and echo length = 15 ms. The images were then analyzed at the workstation provided by GE healthcare.

Supporting Information

Supporting Information is available from the Wiley Online Library or from the author.

Acknowledgements

The authors gratefully acknowledge financial support from the National Natural Science Foundation of China Research (Grant No. 50823007, 50972154, 51132009, 51072212, 51102259, 21172043), the Shanghai Rising-Star Program (Grant No. 12QH1402500), the Science and Technology Commission of Shanghai (Grant No. 11nm0505000, 10430712800), the National Basic Research Program of China (973 Program, Grant No.2011CB707905). The authors thank Gianni Ferrante (STELAR s.r.l., Italy) and Rongrong Shao (Cross-Tech Development CO., Ltd., China) for their kind help in NMRD profile testing; Jing Zhou, Xingjun Zhu, and Prof. Fuyou Li (from Advanced Materials Laboratory, Department of Chemistry, Fudan University) for kind help in UCL in vivo imaging; Jingwei Feng and Lingling Zhang (both from Shanghai Institute of Ceramics, Chinese Academy of Sciences) for TEM characterizations; Yan Yang (Shanghai Institute of Ceramics, Chinese Academy of Sciences) for optical analysis; and Huaiyong Xing and Qingfeng Xiao (both from

Shanghai Institute of Ceramics, Chinese Academy of Sciences) for helpful discussions.

Received: June 1, 2012

Revised: July 25, 2012

Published online:

- [1] F. Wang, Y. Han, C. S. Lim, Y. Lu, J. Wang, J. Xu, H. Chen, C. Zhang, M. Hong, X. Liu, *Nature* **2010**, 463, 1061.
- [2] F. Wang, R. Deng, J. Wang, Q. Wang, Y. Han, H. Zhu, X. Chen, X. Liu, *Nat. Mater.* **2011**, 10, 968.
- [3] R. Kumar, M. Nyk, T. Y. Ohulchanskyy, C. A. Flask, P. N. Prasad, *Adv. Funct. Mater.* **2009**, 19, 853.
- [4] Y. I. Park, J. H. Kim, K. T. Lee, K. S. Jeon, H. Bin Na, J. H. Yu, H. M. Kim, N. Lee, S. H. Choi, S. I. Baik, H. Kim, S. P. Park, B. J. Park, Y. W. Kim, S. H. Lee, S. Y. Yoon, I. C. Song, W. K. Moon, Y. D. Suh, T. Hyeon, *Adv. Mater.* **2009**, 21, 4467.
- [5] H. Guo, Z. Q. Li, H. S. Qian, Y. Hu, I. N. Muhammad, *Nanotechnology* **2010**, 21, 125602.
- [6] J. Zhou, Y. Sun, X. X. Du, L. Q. Xiong, H. Hu, F. Y. Li, *Biomaterials* **2010**, 31, 3287.
- [7] F. Chen, W. Bu, S. Zhang, X. Liu, J. Liu, H. Xing, Q. Xiao, L. Zhou, W. Peng, L. Wang, J. Shi, *Adv. Funct. Mater.* **2011**, 21, 4285.
- [8] N. J. J. Johnson, W. Oakden, G. J. Stanis, R. Scott Prosser, F. C. J. M. van Veggel, *Chem. Mater.* **2011**, 23, 3714.
- [9] J. Zhou, M. Yu, Y. Sun, X. Zhang, X. Zhu, Z. Wu, D. Wu, F. Li, *Biomaterials* **2011**, 32, 1148.
- [10] Q. Ju, D. Tu, Y. Liu, R. Li, H. Zhu, J. Chen, Z. Chen, M. Huang, X. Chen, *J. Am. Chem. Soc.* **2011**, 134, 1323.
- [11] a) M. Haase, H. Schäfer, *Angew. Chem. Int. Ed.* **2011**, 50, 5808; b) X. Xue, F. Wang, X. Liu, *J. Mater. Chem.* **2011**, 21, 13107; c) F. Zhang, G. B. Braun, A. Pallaoro, Y. Zhang, Y. Shi, D. Cui, M. Moskovits, D. Zhao, G. D. Stucky, *Nano Lett.* **2011**, 12, 61; d) J. Zhou, Z. Liu, F. Y. Li, *Chem. Soc. Rev.* **2012**, 41, 1323; e) J. Zhou, X. Zhu, M. Chen, Y. Sun, F. Li, *Biomaterials* **2012**, 33, 6201.
- [12] a) D. Tu, L. Liu, Q. Ju, Y. Liu, H. Zhu, R. Li, X. Chen, *Angew. Chem. Int. Ed.* **2011**, 50, 6306; b) R. Deng, X. Xie, M. Vendrell, Y.-T. Chang, X. Liu, *J. Am. Chem. Soc.* **2011**, 133, 20168.
- [13] C. Wang, H. Tao, L. Cheng, Z. Liu, *Biomaterials* **2011**, 32, 6145.
- [14] D. K. Chatterjee, A. J. Rufal, Y. Zhang, *Biomaterials* **2008**, 29, 937.
- [15] a) Q. Liu, Y. Sun, T. Yang, W. Feng, C. Li, F. Li, *J. Am. Chem. Soc.* **2011**, 133, 17122; b) Y. Yang, Q. Shao, R. Deng, C. Wang, X. Teng, K. Cheng, Z. Cheng, L. Huang, Z. Liu, X. Liu, B. Xing, *Angew. Chem. Int. Ed.* **2012**, 51, 3125.
- [16] S. H. Nam, Y. M. Bae, Y. I. Park, J. H. Kim, H. M. Kim, J. S. Choi, K. T. Lee, T. Hyeon, Y. D. Suh, *Angew. Chem. Int. Ed.* **2011**, 50, 6093.
- [17] S. W. Wu, G. Han, D. J. Milliron, S. Aloni, V. Altoe, D. V. Talapin, B. E. Cohen, P. J. Schuck, *Proc. Natl. Acad. Sci. USA* **2009**, 106, 10917.
- [18] L. Xiong, T. Yang, Y. Yang, C. Xu, F. Li, *Biomaterials* **2010**, 31, 7078.
- [19] Y. Liu, K. Ai, J. Liu, Q. Yuan, Y. He, L. Lu, *Angew. Chem. Int. Ed.* **2012**, 51, 1437.
- [20] P. Caravan, J. J. Ellison, T. J. McMurphy, R. B. Lauffer, *Chem. Rev.* **1999**, 99, 2293.
- [21] S. Aime, M. Botta, M. Fasano, E. Terreno, *Chem. Soc. Rev.* **1998**, 27, 19.
- [22] E. J. Werner, A. Datta, C. J. Jocher, K. N. Raymond, *Angew. Chem. Int. Ed.* **2008**, 47, 8568.
- [23] L. Helm, *Future Med. Chem.* **2010**, 2, 385.
- [24] C. Paquet, H. W. de Haan, D. M. Leek, H.-Y. Lin, B. Xiang, G. Tian, A. Kell, B. Simard, *ACS Nano* **2011**, 5, 3104.
- [25] T. Kim, E. Momin, J. Choi, K. Yuan, H. Zaidi, J. Kim, M. Park, N. Lee, M. T. McMahon, A. Quinones-Hinojosa, J. W. M. Bulte, T. Hyeon, A. A. Gilad, *J. Am. Chem. Soc.* **2011**, 133, 2955.

- [26] Y.-K. Peng, C.-W. Lai, C.-L. Liu, H.-C. Chen, Y.-H. Hsiao, W.-L. Liu, K.-C. Tang, Y. Chi, J.-K. Hsiao, K.-E. Lim, H.-E. Liao, J.-J. Shyue, P.-T. Chou, *ACS Nano* **2011**, 5, 4177.
- [27] E. Toth, R. D. Bolskar, A. Borel, G. Gonzalez, L. Helm, A. E. Merbach, B. Sitharaman, L. J. Wilson, *J. Am. Chem. Soc.* **2005**, 127, 799.
- [28] J. S. Ananta, B. Godin, R. Sethi, L. Moriggi, X. W. Liu, R. E. Serda, R. Krishnamurthy, R. Muthupillai, R. D. Bolskar, L. Helm, M. Ferrari, L. J. Wilson, P. Decuzzi, *Nat. Nanotechnol.* **2010**, 5, 815.
- [29] J. K. Hsiao, C. P. Tsai, T. H. Chung, Y. Hung, M. Yao, H. M. Liu, C. Y. Mou, C. S. Yang, Y. C. Chen, D. M. Huang, *Small* **2008**, 4, 1445.
- [30] K. M. L. Taylor, J. S. Kim, W. J. Rieter, H. An, W. L. Lin, W. B. Lin, *J. Am. Chem. Soc.* **2008**, 130, 2154.
- [31] C. P. Tsai, Y. Hung, Y. H. Chou, D. M. Huang, J. K. Hsiao, C. Chang, Y. C. Chen, C. Y. Mou, *Small* **2008**, 4, 186.
- [32] Z. L. Cheng, D. L. J. Thorek, A. Tsourkas, *Adv. Funct. Mater.* **2009**, 19, 3753.
- [33] W. J. M. Mulder, R. Koole, R. J. Brandwijk, G. Storm, P. T. K. Chin, G. J. Strijkers, C. de Mello Donegá, K. Nicolay, A. W. Griffioen, *Nano Lett.* **2005**, 6, 1.
- [34] J. M. Hooker, A. Datta, M. Botta, K. N. Raymond, M. B. Francis, *Nano Lett.* **2007**, 7, 2207.
- [35] F. Chen, W. B. Bu, Y. Chen, Y. C. Fan, Q. J. He, M. Zhu, X. H. Liu, L. P. Zhou, S. J. Zhang, W. J. Peng, J. L. Shi, *Chem.-Asian J.* **2009**, 4, 1809.
- [36] J. Liu, W. Bu, S. Zhang, F. Chen, H. Xing, L. Pan, L. Zhou, W. Peng, J. Shi, *Chem.-Eur. J.* **2012**, 18, 2335.
- [37] Y. J. Wong, L. Zhu, W. S. Teo, Y. W. Tan, Y. Yang, C. Wang, H. Chen, *J. Am. Chem. Soc.* **2011**, 133, 11422.
- [38] J. Kim, H. S. Kim, N. Lee, T. Kim, H. Kim, T. Yu, I. C. Song, W. K. Moon, T. Hyeon, *Angew. Chem. Int. Ed.* **2008**, 47, 8438.
- [39] N. Bogdan, F. Vetrone, G. A. Ozin, J. A. Capobianco, *Nano Lett.* **2011**, 11, 835.
- [40] M. C. Alpoim, A. M. Urbano, C. F. G. C. Geraldles, J. A. Peters, *J. Chem. Soc., Dalton Trans.* **1992**, 463.
- [41] A. Beeby, I. M. Clarkson, R. S. Dickins, S. Faulkner, D. Parker, L. Royle, A. S. de Sousa, J. A. Gareth Williams, M. Woods, *J. Chem. Soc., Perkin Trans.* **1999**, 2, 493.
- [42] R. M. Supkowski, W. D. Horrocks, *Inorg. Chim. Acta* **2002**, 340, 44.
- [43] K. A. Abel, J.-C. Boyer, C. M. Andrei, F. C. J. M. van Veggel, *J. Phys. Chem. Lett.* **2011**, 2, 185.
- [44] B. H. Kim, N. Lee, H. Kim, K. An, Y. I. Park, Y. Choi, K. Shin, Y. Lee, S. G. Kwon, H. B. Na, J.-G. Park, T.-Y. Ahn, Y.-W. Kim, W. K. Moon, S. H. Choi, T. Hyeon, *J. Am. Chem. Soc.* **2011**, 133, 12624.
- [45] Q. Liu, Y. Sun, C. Li, J. Zhou, C. Li, T. Yang, X. Zhang, T. Yi, D. Wu, F. Li, *ACS Nano* **2011**, 5, 3146.
- [46] L. Xiong, Z. Chen, Q. Tian, T. Cao, C. Xu, F. Li, *Anal. Chem.* **2009**, 81, 8687.
- [47] L. Cheng, K. Yang, M. W. Shao, X. H. Lu, Z. Liu, *Nanomedicine* **2011**, 6, 1327.
- [48] M. L. Matson, L. J. Wilson, *Future Med. Chem.* **2010**, 2, 491.
- [49] P. Hermann, J. Kotek, V. Kubicek, I. Lukes, *Dalton Trans.* **2008**, 3027.
- [50] M. R. Prince, H. L. Zhang, J. C. Prowda, M. E. Grossman, D. N. Silvers, *Radiographics* **2009**, 29, 1565.
- [51] J. Y. Park, M. J. Baek, E. S. Choi, S. Woo, J. H. Kim, T. J. Kim, J. C. Jung, K. S. Chae, Y. Chang, G. H. Lee, *ACS Nano* **2009**, 3, 3663.
- [52] H. Hifumi, S. Yamaoka, A. Tanimoto, D. Citterio, K. Suzuki, *J. Am. Chem. Soc.* **2006**, 128, 15090.
- [53] F. Evanics, P. R. Diamente, F. van Veggel, G. J. Stanisz, R. S. Prosser, *Chem. Mater.* **2006**, 18, 2499.
- [54] L. M. Bronstein, X. L. Huang, J. Retrum, A. Schmucker, M. Pink, B. D. Stein, B. Dragnea, *Chem. Mater.* **2007**, 19, 3624.